

Figure 3–8. BF TEM images of the calcite crystals synthesized without additives (a,b), with soluble OMs extracted from *Pinctada* (c,d) and from *Atrina* (e,f), and with PANa (g,h). (a,c,e,g) were imaged with crystal orientation to form intense diffraction contrast and (b,d,f,h) were imaged with an under-focused condition. Electron diffraction patterns acquired are inserted.

observed in all specimens except the crystals without organic additives, indicating that the Fresnel contrasts correspond to the OMs added (Figure 3–8b, 8d, 8f and 8h). The OMs extracted from *Pinctada* are larger than those from *Atrina* and PANa, showing that they may be likely to form aggregates. Furthermore the shapes of the OMs are not spheres but ellipsoids elongated to particular directions. The elongated directions appear to be along {104} or {001} planes of calcite when the corresponding diffraction patterns are considered, but detailed three-dimensional analysis is needed to conclude it.

3.3.3 STEM–EELS analysis

The Fresnel contrasts in the crystals synthesized with the OMs from *Pinctada* were analyzed also using EELS in the same way as described in Chapter 2A (Figure 3–9). The Fresnel contrasts were recognized as dark contrasts in HAADF–STEM images. Compared with the spectrum obtained from only calcite crystals (black line), that from the dark contrasts in HAADF–STEM images (red line) shows the 289 eV peak with a shoulder at the lower energy-loss side, indicating the existence of the 284 eV peak,



Figure 3–9. EELS spectra acquired from the calcite crystals synthesized with soluble OMs extracted from *Pinctada*. Black spectrum acquired from only calcite crystals, and red one from the dark contrasts in HAADF–STEM images or the Fresnel contrasts in under-focused TEM images. Arrow indicates the location of the 284 eV peak.

namely the organic additives inside the crystals. Thus it is concluded that the Fresnel contrasts in under-focused TEM images correspond to OMs incorporated into crystals.

3.3.4 XRD analysis

XRD analyses were also conducted for the synthesized calcite crystals (Figure 3–10). The crystals precipitated in the solution, not on the SAMs, were used for XRD measurements because the amount of those precipitated on the SAMs was too small. The local lattice strain along the a_i -axes is not distinct among all synthetic calcite crystals because the slopes of the regression lines are similar (Figure 3–10b). On the contrary, the local lattice strain along the *c*-axis is the largest for the crystals with the OMs from *Pinctada*, followed by those with the OMs from *Atrina* and with PANa (Figure 3–10a). This tendency is comparable to the biogenic calcite, although the



Figure 3–10. Williamson–Hall plots of the calcite crystals synthesized in the presence of soluble OMs extracted from *Pinctada* and *Atrina*, and PANa for the 006 and 00.12 (a), 110 and 220 (b) peaks in XRD profiles.

difference between the synthetic specimens is less distinct (see the results in Chapter 2A). These results indicate that the soluble OMs from *Pinctada* can induce more local lattice strain, and consequently make the calcite crystals crystallographically less perfect. This crystallographic imperfection possibly leads to the formation of sub-grains divided by small-angle misorientation in the prisms of *Pinctada*. It is noteworthy that PANa has the least significant influence on calcite crystal structure in spite of its negative charge. All of the y-intercept is smaller than 0.001 Å⁻¹, which corresponds to the coherence length of 94 nm, therefore the coherence length is larger than 94 nm. Since the limit for estimating the coherence length from the peak width of XRD profiles is approximately 100 nm as described in Chapter 2A, the accurate coherence length was not obtained here.

3.3.5 Amino acid composition analysis

Finally, I analyzed the amino acid compositions of the intracrystalline OMs that were extracted from the prisms of *Pinctada* and *Atrina* (Figure 3–11). Since both of the OMs contain a large amount of Asx (Asn and Asp), most proteins in the OMs are probably negatively charged. It is noteworthy that approximately a half of the amino acids (45.6 residue %) in the OMs from *Atrina* is composed of Asx, indicating that the OMs from *Atrina* are much more negatively charged than those from *Pinctada*. In addition, PANa, which was employed as an organic additive in the control synthetic experiments, is definitely a highly acidic polymer.



Figure 3–11. Amino acid compositions of soluble OMs extracted from the prisms of *Pinctada* and *Atrina*.

3.4 Discussion

In vitro synthetic experiments revealed that intracrystalline OMs extracted from the calcite prisms of *Pinctada* interact more strongly with calcite crystals, and consequently induce defective structure and lattice strain. On the other hand, resulting crystals became smaller when the OMs from *Atrina* and PANa were added to the *in vitro* setup, indicating that negatively charged OMs delay crystal growth. However, they have little influence on the crystal structure, which exhibited a perfect single-crystalline feature. These results are comparable with the defect-rich/free structure in the prisms of *Pinctada/Atrina*, implying that soluble OMs are one of the causes for constructing such

microstructures. In addition, the OMs extracted from *Pinctada* can interact more intimately with calcite crystals than more negatively charged OMs such as PANa, which means that other factors than the negative charge are responsible for the interaction.

Recently Kim et al. (2011) also reported that the OMs incorporated into crystals become ellipsoidal shapes. In their study, originally spherical shaped copolymer micelles changed to show oval cross-sections inside the crystals. Such shape change of the OMs occurs probably because the OMs effectively interact with the specific crystallographic planes of calcite. Although the original shapes of OMs added in this study are unknown, they may also interact with some particular planes more strongly because the OMs showed ellipsoidal shapes inside the crystals and were elongated along almost the same directions. Investigating the elongated directions of the OMs may lead to deep understanding of the mechanism by which the OMs are incorporated into crystals.

The defective structure in the prisms of *Pinctada* was reproduced by EDTA-soluble OMs in this study. Such structure has not been produced yet using only insoluble OMs. In many cases, defect-free single crystals were grown in a hydrogel of insoluble OMs owing to reduction of nucleation rate and suppression of ion diffusion. For example, Oaki et al. (2007) formed calcite crystals inside an agar gel, which are composed of apparent sub-units of several tens nanometers but show a complete single-crystalline feature. Furthermore the crystals retain a single-crystalline character even though the amount of insoluble OMs inside crystals become larger by increasing the concentration of the hydrogel. Whereas Grassmann et al. (2003) used a poly-acrylamide gel for producing calcite crystals with microstructures that consist of highly aligned crystallites of submicrometers. The characters of the crystals are similar

to those of microstructures in the prisms of *Pinctada*. Nevertheless the adjacent crystallites do not connect coherently, and the whole crystal holds large orientation spread. Thus insoluble OMs cannot induce microstructures with gradual orientation change in biogenic calcite. The interaction between soluble and insoluble OMs may be essential to create the crystals that are composed of sub-grains of several hundred nanometers divided by small-angle misorientation.

Although soluble OMs do induce defective structure in crystals, the amount of local lattice strain is rather smaller than that in biogenic crystals as shown in Williamson–Hall plots obtained using XRD measurements. Hence the microstructures in biogenic calcite may be reproduced more accurately by mutual influence of soluble and insoluble OMs.

3.5 Conclusions

The defect-rich calcite crystals in the prisms of *Pinctada* were reproduced to some extent *in vitro* by adding the intracrystalline OMs extracted from the prisms, whereas defect-free single crystals were formed even though the OMs from the prisms of *Atrina* and PANa were incorporated into the crystals. Since the OMs from *Atrina* and PANa are more negatively charged than those from *Pinctada*, the negative charge does not directly mean strong affinity for calcite crystals. These results indicate that certain soluble OMs have the responsibility to modify crystal structure and form distinctive microstructures in biominerals, which has been often advocated to date but scarcely demonstrated experimentally. This study suggests that the interaction between soluble OMs and

crystals definitely plays an important role in forming characteristic microstructures in biogenic calcite. Eventually harmonious relationships between soluble/insoluble OMs and crystals generate elaborate structure in biominerals. Chapter 4.

Summary and general conclusions

This study focused on intracrystalline OMs in biominerals and aimed to reveal the interaction between crystals and the OMs. To work toward this goal, I investigated the calcite crystals constituting the outer layers of mollusk shells even at a nanoscale, and deliberated upon the functions of organic phases inside crystalline phases. The visualization of intracrystalline OMs by combining various electron microscopic analyses enabled me to examine the distribution and shapes of the OMs, and their spatial relationships with crystals. The outcome indicated that intracrystalline OMs include the OMs with thin reticulate form and those soluble in EDTA, which individually play distinct roles in biomineralization. The following is the summary of the results obtained in this study.

First, the detailed investigation and comparison of the calcite prismatic layers in *Pinctada* with those in *Atrina* were conducted. The prisms of *Pinctada* contain membranous and net-like OMs, which are distributed apparently inhomogeneously inside the crystals. Small-angle misorientation is generated at the locations of the OMs. As a result, the crystals are composed of sub-grains accompanying gradual orientation change and a large amount of lattice strain. OMs also exist inside the prisms of *Atrina*, but their shapes are thin fibers, which exhibit seeming homogeneous distribution in TEM observation. Since the thin fibrous OMs in *Atrina* have little influence on crystal structure unlike those in *Pinctada*, the prisms are defect-free single crystals.

The calcite crystals constituting the outer layers in *Pteria* and *Crassostrea* of bivalves, and *Haliotis* and *Collisella* of gastropods have similar microstructures to the defect-rich structure in the prisms of *Pinctada*. The crystal defects are induced by reticulate OMs also in these shells, implying that the constitution is a common character in diverse mollusk shells. The gradual orientation change originated from the defects

improves mechanical properties of the biogenic calcite by inhibiting propagation of cleavages along {104} planes. Thus it is reasonably concluded that organisms create defect-rich structure in the skeletons for protecting their soft bodies.

Although it is generally accepted that the reticulate OMs adjust crystal nucleation and growth rates and suppress ion diffusion by supplying environment for crystallization, they do not have influence on crystal structure. In fact, calcite crystals with defective structure have not been duplicated even though the crystals include the reticulate OMs. Whereas many previous studies demonstrated that OMs soluble in acid and/or EDTA are capable of affecting crystal structure. Hence I examined the influence of the OMs extracted by EDTA from the prisms of *Pinctada* and *Atrina* on crystal structure. Intracrystalline OMs extracted from these shells were added to the calcium-containing solution for in vitro crystallization. Defective structure was induced in the resulting calcite crystals that were synthesized in the presence of the EDTA-soluble OMs from Pinctada, but in contrast, those from Atrina did not influence on crystal structure. This synthetic experiment evidenced that EDTA-soluble OMs participate in the formation of defect-rich structure in the prisms of Pinctada. Nevertheless the synthesized crystals hold fairly smaller amount of strain than actual calcite prisms of *Pinctada*. Therefore I assert that the defect-rich structure in biogenic calcite is yielded by mutual influence of both reticulate and EDTA-soluble OMs. Some identified proteins that are included in mollusk shells actually have an ability to bind to the reticulate OMs such as chitin (Suzuki et al., 2004; Suzuki and Nagasawa, 2007; Suzuki et al., 2009).

On the basis of the results obtained in this study, I propose the following mechanism by which intracrystalline OMs regulate the microstructure of calcite crystals

115



Figure 4–1. Schematic illustration of the prismatic structure, showing the mechanism by which intracrystalline OMs regulate the microstructure of calcite crystals constituting outer layers of mollusk shells.

constituting outer layers of mollusk shells (Figure 4-1).

- An organic framework is constructed by OMs which include intercrystalline OMs and subsequent intracrystalline OMs with thin reticulate form. Another kind of OMs soluble in EDTA binds the reticulate OMs.
- As the calcite crystals interact with EDTA-soluble OMs bound to the reticulate OMs, they grow inside the intercrystalline OMs, penetrating the reticulate OMs.

3. These intracrystalline OMs are incorporated into the crystals and build sub-grain microstructures with small-angle misorientation and defect-rich boundaries, which improve the mechanical properties of biogenic calcite by inhibiting propagation of cleavages.

In materials science, for instance, ceramics are stiff but brittle. Nevertheless the brittleness is conquered by decreasing crystallite size and controlling the density of grain boundaries. The grain boundaries block the propagation of cracks across crystallites. Such phenomenon may occur also in the defect-rich structure in biogenic calcite. However, the mechanism for improving the mechanical properties in biogenic calcite must be investigated independently in more detail because the defects are small-angle misorientation generated by the incorporation of OMs into crystals. It is noteworthy here that a lot of energy is needed for the processes of making ceramic materials such as sintering, on the contrary, biominerals are costlessly created at ambient temperature and pressure. Thus understanding the formation mechanism of biominerals gives hints for manufacturing structural materials at low cost.

Future work is to produce crystals with microstructures more similar to biominerals by combining soluble OMs with reticulate OMs in synthetic *in vitro* experiments. *In vitro* crystallization enables us to grasp the whole of the biomineralization processes. Furthermore, it is ultimately necessary to verify whether biominerals are actually formed through the processes as stated above. For this purpose, *in situ* examination such as cryo-electron microscopy should be applied to biomineralization.

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